

Impact of Tunnel-Barrier Strength on Magnetoresistance in Carbon Nanotubes

Caitlin Morgan,^{1,2} Maciej Misiorny,^{3,4,5} Dominik Metten,^{1,2,6} Sebastian Heedt,^{7,2} Thomas Schäpers,^{7,2}
Claus M. Schneider,^{1,2,8} and Carola Meyer^{1,2,9,*}

¹Peter Grünberg Institut (PGI-6), Forschungszentrum Jülich, D-52425 Jülich, Germany

²JARA—Fundamentals of Future Information Technologies, D-52425 Jülich, Germany

³Department of Microtechnology and Nanoscience MC2, Chalmers University of Technology, SE-412 96 Göteborg, Sweden

⁴Peter Grünberg Institut (PGI-2), Forschungszentrum Jülich, D-52 425 Jülich, Germany

⁵Faculty of Physics, Adam Mickiewicz University, PL-61 614 Poznań, Poland

⁶Institut de Physique et Chimie des Matériaux de Strasbourg and NIE, Université de Strasbourg and CNRS, UMR 7504 Strasbourg, France

⁷Peter Grünberg Institut (PGI-9), Forschungszentrum Jülich, D-52425 Jülich, Germany

⁸Fakultät für Physik and Center for Nanointegration Duisburg-Essen (CeNIDE), Universität Duisburg-Essen, D-47048 Duisburg, Germany

⁹Fachbereich Physik, Universität Osnabrück, D-49069 Osnabrück, Germany

(Received 10 November 2015; revised manuscript received 16 March 2016; published 17 May 2016)

We investigate magnetoresistance in spin valves involving CoPd-contacted carbon nanotubes. Both the temperature and bias-voltage dependence clearly indicate tunneling magnetoresistance as the origin. We show that this effect is significantly affected by the tunnel-barrier strength, which appears to be one reason for the variation between devices previously detected in similar structures. Modeling the data by means of the scattering matrix approach, we find a nontrivial dependence of the magnetoresistance on the barrier strength. Furthermore, an analysis of the spin precession observed in a nonlocal Hanle measurement yields a spin lifetime of $\tau_s = 1.1$ ns, a value comparable with those found in silicon- or graphene-based spin-valve devices.

DOI: 10.1103/PhysRevApplied.5.054010

I. INTRODUCTION

Low-dimensional carbon systems, such as graphene and carbon nanotubes (CNTs), are envisaged as promising candidates for electronic devices beyond the conventional CMOS technology. In the field of spintronics [1], considerable interest in such systems stems from their inherent properties: long coherence times manifesting as ballistic transport of electrons and long spin lifetimes due to low spin-orbit coupling and hyperfine interaction that can even be zero, if ^{12}C is used in the synthesis. Spin injection into two-dimensional graphene is a well-understood phenomenon, and it is experimentally established that reliable contacts are formed with MgO or Al_2O_3 as the tunnel barrier [2–4]. However, there was a strong preceding debate about the role of the insulator with regard to spin injection [5,6], and ongoing research strives to improve the tunnel barrier even further [7].

Understanding the process of spin injection in its one-dimensional counterpart proves to be far more difficult, mainly due to two reasons. First, it is difficult to reliably fabricate low-resistive ferromagnetic contacts to CNTs. As a result, a substantial variation of the magnitude of the magnetoresistance (MR) effect between different devices

has been observed from early on [8]. Second, the underlying physics of the MR in CNTs is more complex, since the transport regime is determined by the strength of the tunnel coupling between a CNT and the leads. The two limiting cases are the quantum-dot regime, when the coupling is *weak*, and the Fabry-Pérot regime for *strong* coupling leading to high transmission [9]. In the former case, huge variations of the MR effect between -80% and $+120\%$ are observed [10–13]. These occur owing to changes of the position and the width of the Coulomb resonances, which, in turn, depend on the magnetic configuration (parallel or antiparallel) of the magnetic moments of the contacts and oscillate with gate and bias voltage [13]. The oscillating behavior with gate and bias voltage is also found in the strong-coupling regime [14,15], though with a significantly reduced MR amplitude not exceeding 9%.

In the present article, we focus on investigating the barrier dependence of the MR effect in CNT spin valves with low-resistive contacts. The focus is on devices that exhibit metallic conductance. Therefore, the devices are built from double-walled (DW) CNTs, since they have a larger probability of being metallic compared to single-walled (SW) CNTs [16,17]. We show that one reason for the sample-dependent variation of the MR in CNTs is a difference in the strength of the tunnel barrier that forms

*carola.meyer@uos.de

intrinsically between a contact and a CNT. Depending on the barrier, the MR change can be increased above 10% even with simple binary alloys used as the contact material (e.g., CoPd, NiPd, etc.). The identification and optimization of the barrier are therefore of crucial importance, in order to fabricate reliable CNT spin valves with large MR effects. Moreover, we demonstrate a Hanle measurement on a CNT spin valve that not only corroborates the feasibility of the spin injection in this case but also shows that such a system competes with state-of-the-art devices based on graphene and Si in terms of the spin lifetime and spin accumulation.

The paper is organized as follows: In Sec. II, experimental details of the device fabrication are discussed, as comparability and reproducibility of the devices are crucial for this study. To compare the MR effect of different devices, the measurements have to be performed within the same transport regime as explained in Sec. III A. Next, in Sec. III B, we present the experimental results on how the MR effect depends on the tunnel-barrier strength and introduce the theoretical model to support the interpretation of the data. Finally, spin precession within CNTs is investigated by means of the Hanle effect in Sec. III C.

II. EXPERIMENTAL DETAILS

All devices discussed in this work are fabricated on heavily *p*-doped Si(001) substrates with a 200-nm layer of thermally oxidized SiO₂. Nanotubes are subsequently grown from patterned islands by chemical vapor deposition (CVD) with an iron-based catalyst [18]. The contacts are patterned by using electron-beam lithography. Recently, permalloy and ferromagnet-Pd alloys have shown promising results [14,19,20]. At present, we focus our research on CoPd, which is demonstrated to exhibit a large in-plane magnetization and can form low-resistive contacts to CNTs [21,22]. Co and Pd are coevaporated via molecular beam epitaxy to create nanofabricated contacts, while Au is deposited via standard metal evaporation to provide coarser leads. A typical resulting lateral spin-valve structure is shown in the inset in Fig. 1. The distance between the CoPd contacts is 150 nm, and the width of the contacts is 150 and 400 nm, respectively, which in turn leads to different aspect ratios for the two contacts and in consequence to different coercive fields due to shape anisotropy. A non-trivial temperature dependence of the coercive field is found and attributed to the interplay of the shape anisotropy and the magnetoelastic effect [22]. The influence of this effect on the switching in magnetoresistance measurements is analyzed and discussed below (see Sec. III A).

The CVD process results in CNTs of mixed chirality, so that both semiconducting as well as metallic tubes are, in general, expected to be present. Since the aim is to compare similar devices, we choose a growth temperature of 920 °C that yields mainly double-walled CNTs [23] and, thus, enhances the probability of fabricating metallic devices. All devices compared in this work show linear current-voltage

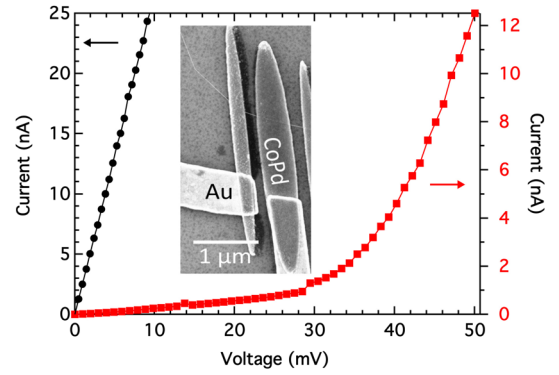


FIG. 1. Current-voltage characteristics of a typical device under consideration, which exhibits metallic behavior at room temperature (black circles), whereas the formation of a potential barrier at 4 K (red squares) is observed. The inset shows an image of a typical device taken by a scanning electron microscope. The position of the tube has been redrawn to enhance visibility.

characteristics at room temperature, while at 4 K a potential barrier becomes apparent, as one can see in Fig. 1. Note that, while these data correspond to one sample (device 6; cf. Table I), they are in fact representative for all devices reported, although the height of the potential barrier varies between different devices. This intrinsic tunnel barrier is crucial for measuring a magnetoresistance effect in CNTs, because otherwise the spin polarization would be lost due to the conductivity mismatch [24].

The interface between the ferromagnetic contact and the CNT is crucial for transport applications. To begin with, we note that the contact interface is quite small, which is a consequence of the fact that DW CNTs have diameters of a few nanometers (cf. Table I for the diameters of the CNTs in the devices presented in this paper). Furthermore, since the injection of charge carriers occurs only at the point where the contact ends [25], the local magnetic environment has a strong influence on the MR measurements. Importantly, the pinning and moving of domain walls in contacts that incorporate several magnetic domains leads to unstable switching behavior of the contacts and a nontrivial MR signal [26]. For this reason, we optimize the shape of the contacts in such a way that the magnetization is in-plane along the long axis with a single or at most two domains [22]. The magnetic field in all magnetoresistance measurements presented in this work is applied in the direction of the long axis of the contacts unless specified otherwise.

All measurements discussed here are performed at a temperature of $T = 4$ K unless specified otherwise. The substrate serves as a back gate, and the back-gate voltage is kept constant for all measurements on the same device, as the gate voltage can strongly influence the MR effect [14,15]. In order to be able to compare the magnitude of the MR effect of different devices, measurements presented here are carried out in a regime where the current showed no, or only very weak, dependence on the back-gate voltage.

III. RESULTS AND DISCUSSION

A. Temperature and bias dependence of the magnetoresistance

Figure 2 shows typical MR measurements taken on a CoPd-contacted CNT in a temperature range from 3.4 up to 75 K with a constant bias $V_{\text{bias}} = 2$ mV. For the sake of clarity, results for downward and upward field sweeps are plotted separately in Figs. 2(a) and 2(b). The data clearly reveal switching between low- and high-resistive states for parallel and antiparallel configurations of the magnetization directions of the contacts. As for all devices presented in this work, the MR effect is positive under the given circumstances. The overall resistance is shown to decrease with rising temperature, as expected for electrons tunneling through a barrier [27], resulting in the offset observed between the curves. The MR signal decreases in magnitude along with the resistance, all in agreement with previous measurements on Fe-contacted CNTs [11]. Along with the amplitude of the MR switching, the width of the feature is also reduced with increasing temperature from $\Delta H_c = \pm 13$ mT at 3.4 K to $\Delta H_c \leq 4$ mT at 50 K. The width of the MR signal is determined by shape anisotropy, as the contacts are fabricated to have different lateral dimensions. The observed temperature-dependent change in the width of the MR signal corresponds to a changing difference in the coercive fields of the contacts. This behavior, in turn, correlates well with the

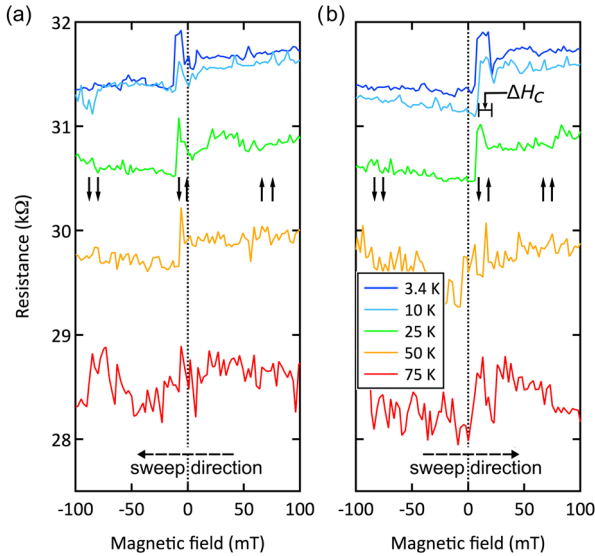


FIG. 2. The dependence of local magnetoresistance curves on temperature investigated in a temperature range of 3.4–75 K at $V_{\text{bias}} = 2$ mV. The left (right) panel shows results for a field sweep in the negative (positive) direction as indicated by the dashed arrows. The presented data set corresponds to device 2 (cf. Table I). Solid arrows represent the magnetization direction of the two contacts.

temperature-dependent SQUID data of CoPd [22], which shows that, at 50 K, magnetoelastic anisotropy strongly affects the magnetization of nanocontact arrays, resulting in a lowered coercive field. This effect is different for the two contacts because of the interplay of shape anisotropy and magnetoelastic anisotropy. The reduced magnitude of the MR signal at elevated temperatures cannot be attributed to a decrease in the magnetization of the CoPd contacts, since their saturation magnetization is constant within this temperature range [22]. Therefore, this is an indication that the MR effect is due to tunneling, and we refer to it as tunneling magnetoresistance (TMR) in the following.

Figures 3(a) and 3(b) show MR measurements of another device at 4 K using different bias voltages V_{bias} . Though the background of the traces and the details of the TMR switching vary from sweep to sweep and, in particular, between different thermal cycles, the coercive fields and the size of the MR effect observed remain the same.

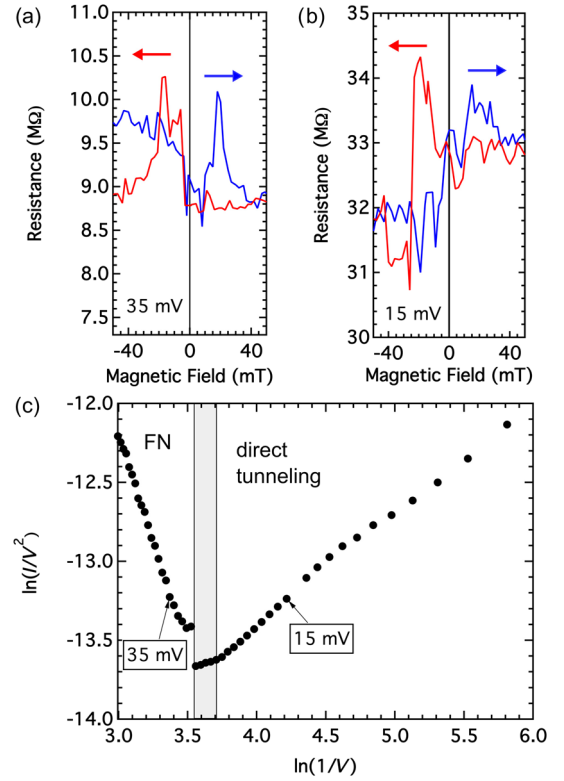


FIG. 3. Magnetoresistance measurements of the sample presented in Fig. 1 obtained at bias voltage: (a) $V_{\text{bias}} = 35$ mV with TMR = 15% and (b) $V_{\text{bias}} = 15$ mV with TMR = 6%. Arrows indicate the sweep direction of the magnetic field. (c) The $\ln(I/V^2) - \ln(1/V)$ plot of the current-voltage characteristics at $T = 4$ K and with $B = 0$ T reveals that the different sizes of the MR effect can be attributed to different tunneling regimes. The gray shaded region marks the transition regime between direct tunneling (right) and Fowler-Nordheim tunneling (left) that is targeted for the following measurements.

Introducing the definition of the size of the tunneling magnetoresistance effect

$$\text{TMR} = \frac{G_P - G_{AP}}{G_{AP}}, \quad (1)$$

with G_P (G_{AP}) referring to the conductance in the parallel (antiparallel) magnetic configuration of the contacts, and averaging the backgrounds, we obtain $\text{TMR} = 15\%$ for the measurement with bias voltage $V_{\text{bias}} = 35$ mV [Fig. 3(a)] and $\text{TMR} = 6\%$ for the measurement with $V_{\text{bias}} = 15$ mV [Fig. 3(b)]. At first glance, this seems to contradict the general observation that the size of the MR effect in CNTs decreases with increasing bias voltage [10,15]. Interestingly, the $\ln(I/V^2) - \ln(1/V)$ plot shown in Fig. 3(c) reveals that the two measurements correspond to different tunneling regimes: The low-bias measurement corresponds to the *direct* tunneling regime, while the high-bias measurement corresponds to the Fowler-Nordheim (FN) tunneling regime [28]. In the direct tunneling regime, the bias voltage applied is smaller than the average barrier height. On the other hand, in the case of FN tunneling, the bias voltage exceeds the average barrier height. As a consequence, the effective barrier width is reduced and the probability of tunneling enhanced. Usually, a $\ln(I/V^2) - 1/V$ plot is used to analyze FN tunneling through an oxide barrier [29]. However, the double-logarithmic plot serves to emphasize the transition between the tunneling regimes as the change of slope becomes clearly visible [30]. Note that we do observe oscillations of the conductance with gate voltage at low bias voltage $V \rightarrow 0$ mV. In contrast to Man, Wever, and Morpurgo [15], the data in this work are acquired at an elevated bias voltage, where we do not observe strong oscillations of the conductance. The fact that we work with larger bias voltages is owed to comparably stronger tunnel barriers, which might also explain why in Ref. [15] the MR signal vanishes at a bias voltage of 20 mV while our measurements show a large TMR signal at similar bias voltages.

A decreasing MR effect with increasing bias voltage is typical for conventional magnetic tunnel junctions and organic spin-valve structures in the direct tunneling regime [31–34]. Spin-selective processes can play a role in the FN tunneling regime due to either band alignment [35] or trap-assisted tunneling [36] and lead to an increase in TMR [30,37]. Before we can understand how these effects might be influenced by the tunnel barrier between the metal contact and a carbon nanotube, we need to find a way to reliably characterize and understand the general influence of this tunnel barrier on the MR effect in CNTs. To exclude an influence of the bias voltage on the signal, we compare the TMR values of different CNT devices measured in the very narrow transition region between these two tunneling regimes, as represented by the gray shaded area in Fig. 3(c). This, of course, results in different absolute values of the applied bias voltage, since the conductances of the investigated devices differ. The absolute current through all the devices within this regime, however, is the same with $I = 1$ nA.

B. Dependence of magnetoresistance on the strength of the tunnel barrier

Table I lists the devices used for TMR measurements, and it reveals that these devices differ significantly with respect to the tube diameter and resistance. First of all, one should note that Pd contacts generally possess a high transmission, which is also known to increase with the diameter of the nanotube even for metallic CNTs [38].

Indeed, the CoPd-contacted devices under investigation exhibit a similar behavior. Devices 1 and 2, characterized by the largest CNT diameters, show a high transmission, though the theoretical minimal value of the resistance $1/G = h/4e^2$ for a single-walled CNT is not reached. A general trend can be observed that the value of the resistance at room temperature is inversely proportional to the CNT diameter. However, we note that the resistances of devices 5 and 6 appear to be too high compared to

TABLE I. Summary of CNT diameters and electronic transport data for the CoPd-contacted CNT devices discussed in this work. In the case of non-Ohmic current-voltage characteristics at 4 K, the resistance listed is determined for the bias voltage used in the transport measurements. Note that the TMR [see Eq. (1)] is determined at $T = 4$ K. The error for the TMR value as well as for the barrier strength Z [see Eq. (2)] is given for the last digit in brackets. The error for the barrier strength is due to an underestimation of the barrier caused by the temperature dependence of the conductance between room temperature and the temperature where the tunnel barrier forms. Though $1/G(T)$ data exist for all devices, in some cases they are taken at a different bias voltage than the TMR measurement.

Device	CNT diameter (nm)	$1/G(300 \text{ K} \equiv \text{RT})$ (k Ω)	$1/G(4 \text{ K}) \equiv 1/G_p$ (k Ω)	TMR (%)	Z
1	5	35	30	0(0.5)	−0.1(+3)
2	8	25	32	2(0.5)	0.3(+4)
3	1.6	150	210	5(0.75)	0.40(+6)
4	0.7	3700	11 000	6.4(0.5)	1.972(+3)
5	3.2	1000	9000	12(2)	8.00(+1)
6	3	350	100 000	15(1)	24.71(+3)

device 3. This deviation can be explained regarding the fact that most of the CNTs used in the present study are DW or even multiwalled, with device 4 being the only exception of a SW CNT. Though devices exhibiting metallic behavior were specifically selected, one should remember that DW CNTs can also be composed of outer semiconducting CNTs, with the inner tubes being metallic. Such a situation would manifest itself in a higher contact resistance and, in particular, as a higher device resistance in the case of our two-terminal device. Moreover, the strong increase in resistance with decreasing temperature supports this argument, as such a device should form a rather strong tunnel barrier. At a low temperature, individual CNTs exhibit intrinsic tunnel barriers of different strength to metal contacts [39], and this process depends largely on their chirality. Electrons experience a potential barrier at the contact-CNT interface as a result of local hybridization and a local dipolar moment [40,41].

Figure 4 shows how the magnitude of the TMR signal changes with respect to the strength of the tunnel barrier formed between the CNT and the CoPd contacts. We define the *dimensionless effective barrier strength* Z of a device by directly comparing the resistance $1/G$ at 4 K and at room temperature (RT) as follows (see also rightmost column of Table I):

$$Z = \frac{1/G(T) - 1/G(\text{RT})}{1/G(\text{RT})} = \exp\left(\frac{E_b}{k_B T}\right) - 1 \approx \frac{E_b}{k_B T}, \quad (2)$$

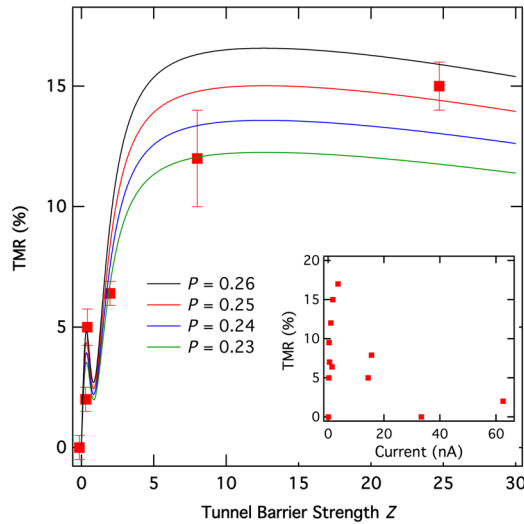


FIG. 4. Evolution of TMR as a function of the tunnel-barrier strength Z at $T = 4$ K. Experimental data (red squares) corresponding to different devices (see Table I) are accompanied by theoretical calculations (solid lines) for several values of the magnetic polarization parameter P of injected charge carriers at fixed gate voltage ($E_g = 6$ meV) based on the model explained in the text. Error bars represent deviations in TMR between multiple measurements. The inset shows the magnitude of TMR with respect to the current through the different devices presented

with $T = 4$ K. Here, we assume that the change in resistance $1/G$ with temperature is dominated by tunnel barriers that are overcome at an elevated temperature ($T > 4$ K) by thermal activation of the charge carriers. This is valid, since all devices investigated exhibit relatively weak temperature dependence during cooldown before the tunnel barrier becomes visible at $T \sim 50$ K. Physically, the dimensionless barrier strength Z describes the relation between the energy E_b required to overcome the potential barrier and the kinetic (thermal) energy of electrons incident at the contact-CNT interface. We note that when an electron traverses the device it actually encounters two barriers, that is, when it enters and leaves a CNT. These are, however, experimentally indistinguishable, and Z includes the overall effect of both tunnel barriers, though essentially Z is determined by the larger one. The normalization with respect to room-temperature resistance allows for a direct comparison with the definition of the barrier strength in the model used to interpret our data (see below).

The data indicate that the tunnel-barrier strength Z does indeed influence the magnitude of the TMR as suggested by Slonczewski for conventional TMR [42], with devices characterized by low barriers unable to achieve an effect larger than a few percent. Once the barrier reaches a certain strength, a further increase will no longer affect the TMR. The inset in Fig. 4 shows a plot of the TMR vs the current of the same devices that is usually used to analyze the performance of spin valves. Since this type of plot does not account for different tunnel barriers and bias-voltage regimes, it is clear that for our devices a general trend cannot be expected.

To support our interpretation of the dependence of the TMR on the tunnel-barrier strength, we compare our results to model calculations of the TMR for a CNT, approximated as a ballistic and noninteracting one-dimensional (gated) quantum wire [43–45] interconnecting two ferromagnetic leads. In essence, such a model corresponds to an electronic interferometer studied previously both in experiment [14,46,47] and theoretically [45,48]. Here, the key element of the model, which has not been addressed so far in full detail, is the effect of the strength of tunnel barriers occurring at the CNT-lead interfaces on spin-dependent transport through the device. As the exact shape of each of the two tunnel barriers is unknown, we describe scattering of tunneling electrons at the CNT-lead interface by means of a repulsive Dirac-delta potential $U_0\delta(x)$, the same at both ends of the CNT. This approach is already proven sufficient to capture some key transport features of the interface for other material systems [49–52].

In general, within the scattering matrix approach the expression for linear-response conductance of a device at temperature T takes the form [53,54]

$$G_{P/AP} = \frac{e^2}{h} \sum_{n,\sigma} \int d\epsilon \mathcal{J}_{n,\sigma}^{P/AP}(\epsilon) \left(-\frac{\partial f(\epsilon)}{\partial \epsilon} \right), \quad (3)$$

where $\mathcal{J}_{n,\sigma}^{P/AP}(\varepsilon)$ denotes the transmission coefficient for an electron of spin σ traversing a device through its n th channel in the case when the spin moments of the leads are oriented parallel (P) or antiparallel (AP) and $f(\varepsilon) = \{1 + \exp[\varepsilon/(k_B T)]\}^{-1}$ is the Fermi-Dirac distribution. Furthermore, we limit our considerations to the case of a CNT, which can support only two orbital channels ($n = 2$)—a generic property inherited from graphene which stems from the presence of two inequivalent carbon atoms in a primitive cell [43,55]. Without losing generality, as long as one assumes two identical and uncoupled channels, a CNT can be approximated as a one-dimensional wire with $\mathcal{J}_{n,\sigma}^c(\varepsilon) = \mathcal{J}_\sigma^c(\varepsilon)$ for $c = P, AP$ [45]. Next, the transmission coefficient $\mathcal{J}_\sigma^c(\varepsilon)$ is derived in a standard manner by means of the scattering matrix approach [48,53,56]:

$$\mathcal{J}_\sigma^c(\varepsilon) = \frac{\mathcal{T}_\sigma \mathcal{T}_{c(\sigma)}}{|1 - \sqrt{(1 - \mathcal{T}_\sigma)(1 - \mathcal{T}_{c(\sigma)})} \exp[i\Phi_\sigma^c(\varepsilon)]|^2}, \quad (4)$$

with \mathcal{T}_σ representing the square of the absolute value of the complex transmission amplitude. Note that when writing the equation above we assume that both leads and interfaces (represented by the parameter U_0) are identical. The action of the magnetic configuration index c on a spin index σ is defined as $P(\sigma) = \sigma$ and $AP(\sigma) = \bar{\sigma}$, where one should understand the notation $\bar{\sigma}$ as $\bar{\uparrow} \equiv \downarrow$ and $\bar{\downarrow} \equiv \uparrow$. In the absence of a magnetic field, the phase factor $\Phi_\sigma^c(\varepsilon) = 2\delta(\varepsilon) + \varphi_\sigma + \varphi_{c(\sigma)}$ consists of two terms: first, $\delta(\varepsilon) = \ell k^w(\varepsilon)$, corresponding to the quantum-mechanical phase acquired by a free electron propagating with a wave vector $k^w(\varepsilon)$ in a wire of length ℓ , and, second, $\varphi_\sigma + \varphi_{c(\sigma)}$, representing the spin-dependent phase shift [48] gained by an electron when it is scattered at the interface back into the wire. If the electrostatic potential energy $|E_g|$ of the gated wire is small relative to its Fermi energy E_F^w , one gets $k^w(\varepsilon) = k_F^w + (\varepsilon - E_g)/(\hbar v_F^w)$, where energy ε is defined relative to the Fermi level, $k_F^w = 8.5 \times 10^9 \text{ m}^{-1}$ is a typical value of the Fermi wave vector for electrons in a SW CNT [46], and $v_F^w = 8 \times 10^5 \text{ m/s}$ is its Fermi velocity.

Using standard quantum-mechanical methods and noting that in the low bias-voltage limit only electrons in the vicinity of the Fermi level of the contacts participate in tunneling, one obtains [57]

$$\mathcal{T}_\sigma = \left| \frac{2\sqrt{k_\sigma k^w(0)}}{k_\sigma + k^w(0) + iZ_0\sqrt{k^w(0)\kappa}} \right|^2 \quad (5)$$

and

$$\varphi_\sigma = \arg \left(\frac{-k_\sigma + k^w(0) - iZ_0\sqrt{k^w(0)\kappa}}{k_\sigma + k^w(0) + iZ_0\sqrt{k^w(0)\kappa}} \right). \quad (6)$$

Here, the spin-dependent wave vector of free electrons in the leads, $k_{\uparrow(\downarrow)} = \sqrt{2m_e E_F(1 \pm 2P)}/\hbar$, is parameterized

by the Fermi energy E_F (assumed here $E_F = 8.5 \text{ eV}$) of the contacts, and the *spin-polarization coefficient* defined as [50] $P = (k_\uparrow - k_\downarrow)/(k_\uparrow + k_\downarrow)$, with $P < 0.5$. Furthermore, $\kappa = 2E_F/(\hbar v_F^w)$ and $Z_0 = k_0 U_0/E_F$ with $k_0 = \sqrt{2m_e E_F}/\hbar$ is the *dimensionless barrier strength*, defined as the ratio of the potential energy of the barrier $k_0 U_0$ and the energy of incident electrons from the Fermi level of the contacts, so that it is related to the experimental definition of the barrier strength [Eq. (2)] as $Z = Z_0$. In consequence, it can be noticed that the only two free parameters of the model are E_g and P .

The fit of the theoretical model to the experimental data points for a specific value of E_g and different degrees of spin polarization P of the contacts is shown as solid lines in Fig. 4. The value of E_g is chosen so that it matches the experimental finding of only positive values for the MR effect for small Z . It is to some extent arbitrary, since the MR shows an oscillatory behavior with E_g . Moreover, the temperature used for the fit is $T = 3.5 \text{ K}$, and the length of the CNT is taken as $\ell = 130 \text{ nm}$. Though both these values deviate slightly from those observed in the experiment, they yield better fits. Nevertheless, this seems reasonable, since, first, lithography of various devices is usually not exact, so that it is expected that they are characterized by a somewhat different CNT length, and, second, for a given value of E_g and large Z , the model is sensitive to the interplay of values of temperature and length [57].

Comparing the results of the calculations and the experimental data, we find that the spin polarization of the charge carriers injected from the CoPd contacts is $24 \pm 2\%$. This is rather large compared to CNTs contacted with NiPd, where the spin polarization is estimated to be $\approx 10\%$ [15]. Nevertheless, this is a very reasonable value considering the fact that the polarization of charge carriers injected from pure Co is found to be between 21% and 35% [58] combined with the relatively large polarization of Pd induced by Co [59]. Furthermore, the model is based on a single transport channel, while most of the CNTs used in the devices contain two walls and thus likely more transport channels. While the qualitative effect remains unchanged if more channels are included in the model [57], this might result in a deviation of the contact polarization. On the other hand, interaction between a CNT and contacts can block channels in a CNT with more than one wall [60] and cancel contributions from other channels. The good agreement of the experimental data with the model further proves the importance of the tunnel-barrier strength in optimizing TMR in a CNT-based device.

The fact that the nature and the strength of the barrier significantly affect the TMR signal is observed in “classical” metallic tunnel junctions (MTJs) with MgO as the insulating barrier. Generally, the TMR effect increases with increasing MgO thickness [61]. However, pronounced oscillations are observed in MTJs with Heusler alloys [62,63] or Fe [61] as the contact material. This is

usually attributed to different wave vectors k of the Δ_1 and the Δ_5 band of MgO at the Fermi level that lead to interference effects [64]. Investigations of graphene spin valves, on the other hand, suggest a monotonic increase of the MR and saturation for strong barriers [4]. Moreover, the rather small TMR signal previously found in CNT spin valves with permalloy contacts and the small contact polarization deduced by using Juli re's model [65] might in fact be explained with a weak tunnel barrier in the framework of our model.

Finally, it should be pointed out that, although the proposed model describes the experimental results relatively well, one should still bear in mind its limitations. The model assumes free (s -band) electrons in the contacts, and the CNT-contact interface is approximated by a Dirac-delta potential. In reality, the energy barrier at the interface can be much more complex, with a potential profile depending on many factors, such as the interface roughness or adsorbates. Also, the free-electron model should be applied to transition metals and their alloys with great care, as in principle one should expect a more complicated band structure to be responsible for electron tunneling [66]. To account for all these details, a model from first principles should be developed [67], a task which is way beyond the scope of the present paper.

Within these limitations, it is nevertheless obvious that the tunnel barrier has a significant effect on the TMR found in CNT spin valves. Furthermore, CoPd-CNT devices form sufficiently low barriers that the addition of tunnel barriers from spin-selective insulators becomes feasible. This would also enable one to study the FN tunneling regime more closely, which may lead to significantly larger TMR values.

C. Spin precession within carbon nanotubes

As the results discussed above correspond to local measurements, there is the possibility of several spurious effects, such as the Hall effect, anisotropic magnetoresistance (AMR) [68], or other Ohmic effects [47] influencing the magnetotransport. One important contribution to the local MR signal can be caused by the magneto-Coulomb effect [69–71]. However, the measurements presented in this work are not taken in the Coulomb blockade regime. For this reason, the influence of the magneto-Coulomb effect should be weak at least. Another effect is the tunneling anisotropic magnetoresistance (TAMR), which can completely mimic the MR effect [72]. TAMR is an effect solely related to the anisotropic density of states in the ferromagnetic contacts and is usually found in materials with strong spin-orbit coupling that lack inversion symmetry and show AMR, like lanthanum strontium manganite [73]. Though we measure no significant AMR in our CoPd contacts, TAMR might be related to the barrier, as shown for fcc Co(111) contacts on Al_2O_3 [74]. Since our contacts have a similar structure [22] and the exact nature of the

barrier in our devices is unknown, we have to make sure that spin-polarized charge carriers are indeed injected into the CNTs.

To prove spin injection into and spin precession within the CNTs, Hanle measurements are performed on a device with a nonlocal three-terminal configuration. Figure 5(a) presents an atomic force microscope image (AFM) of a device prepared for nonlocal measurements, in which a nonlocal voltage is measured along a separate path from the applied current, using three contacts as indicated. The device displays a local TMR signal of 15% and, thus, is an example for the strong barrier case as discussed above. For the Hanle measurements, all contacts are magnetized in-plane in the same direction. The sample is then rotated at zero applied field. Subsequently, a magnetic field is applied perpendicular to the plane, which leaves the magnetization of the contacts unaffected due to the shape anisotropy [22]. The resulting Hanle signal is presented in Fig. 5(b). Note that this signal is recorded at a bias of $I_{\text{bias}} = 10$ nA and thus outside the Coulomb blockade regime. The voltage difference $\Delta V_{\text{nl}}(B)$ measured between the injector and the detector contact at $B = 0$ is due to an imbalance of the electrochemical potentials μ^\uparrow and μ^\downarrow of spin-up and spin-down electrons, respectively. It is reduced as the spins dephase while precessing about the perpendicular magnetic field with the Larmor frequency $\omega_L = g\mu_B B/\hbar$. The data are fitted with a Lorentzian peak (blue line), and the magnetic field at the full width at half maximum of the peak is measured to be approximately 5.2 mT. Following Ref. [75], $\Delta V_{\text{nl}}(B) = \Delta V_{\text{nl}}(0)/[1 + (\omega_L \tau_s)^2]$, where τ_s is the spin lifetime. Taking the g factor $g = 2$ for a CNT in a

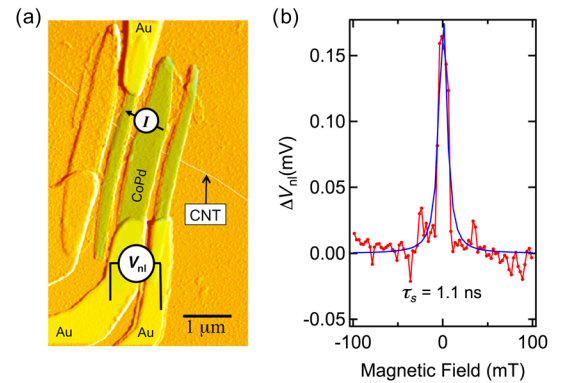


FIG. 5. (a) AFM image of the three-terminal device used in a nonlocal configuration. The CoPd terminals are shaded in green; the gold coarse lines are shaded yellow. The position of the tube has been redrawn for better visibility. The current flow and voltage measurement of the nonlocal signal are schematically indicated. The terminal on the left does not switch its magnetization up to fields as large as 2 T and is left floating. (b) The Hanle measurement (red dots connected by a line that serves merely as a guide to the eyes) is taken at a bias of $I_{\text{bias}} = 10$ nA. The fit (blue) with a Lorentz curve yields the spin lifetime of $\tau_s = 1.1$ ns.

perpendicularly applied magnetic field [76], we obtain $\tau_s = 1.1$ ns. This is a lower bound for the spin-coherence time T_2 , since the Hanle effect in combination with spin injection probes the (coherent) dephasing of the conduction electrons [1]. The value of $\Delta V_{\text{nl}}(0)$ is directly related to the spin accumulation $\Delta\mu(0) = \mu^\uparrow - \mu^\downarrow = 2e\Delta V_{\text{nl}}(0)/\text{TSP}$, with TSP denoting the tunneling spin polarization of the ferromagnet-barrier interface of 0.24 as obtained in the previous section, which yields $\Delta\mu(0) \approx 1.45$ meV. This value is rather large and of similar magnitude to the one found in Si-based spin valves at room temperature [75]. The spin lifetime is long compared to the spin lifetime of 0.1–0.2 ns typically found in graphene lateral spin valves on substrates [77,78]. Only very recently, lifetimes of nanoseconds are reported in high-mobility graphene devices sandwiched in hBN [79] and in suspended graphene devices covered with a layer of BN [3].

As discussed above, our CNT lateral spin valves compete very well with state-of-the-art silicon- or graphene-based devices. Because the performance of devices considered in this paper relies on the tunnel barrier that forms intrinsically between the metal contact and the CNT, the measurements have to be conducted at low temperatures. On the other hand, the spin lifetime in graphene devices is known to exhibit no, or only very little, temperature dependence [78]. Since we now understand the influence of the tunnel barrier in CNT-based devices, tunnel-barrier engineering becomes possible. Spin-selective insulators like EuO can be used to increase the temperature for spin injection. The spin lifetime is very likely still influenced by charge traps in the SiO₂. Suspending the CNTs or low-dose gamma irradiation, which is known to remove traps in the oxide and to improve the performance of CNT-based field-effect transistors [80], would be expected to enhance the spin lifetime. The diameter of the CNT used for the Hanle measurements is 1.5 nm, giving rise to a spin-orbit coupling strength on the order of 1 meV [81]. Since spin-orbit coupling is another source of decoherence, using CNTs with a larger diameter might increase the spin lifetime as well. Recently, a coherence time of 60 ns was found in a CNT-based double quantum dot coherently coupled to microwave cavity photons [82] that indicates the potential of the spin coherence in CNT spin valves. However, electron spin resonance suggests that the electron-electron interactions in a Tomonaga-Luttinger liquid might severely reduce the spin-coherence time of one-dimensional metallic devices [83]. Hanle measurements on improved devices will advance the quantitative understanding of the spin-relaxation properties of CNTs.

IV. CONCLUSION

In this article, we analyze the TMR effect in lateral spin-valve devices based on CNTs. We show that the strength of the tunnel barrier forming intrinsically between ferromagnetic contacts made out of CoPd and CNTs has a significant

influence on the size of the TMR signal. Approximating a CNT as a ballistic and noninteracting one-dimensional quantum wire, we model the experimental data and find that the spin polarization of the injected electrons is about 24% and that the TMR effect exhibits a nonmonotonic dependence on the barrier strength. Importantly, the usage of CoPd as the contact material allows for achieving rather high transmission coefficients, with great perspectives for further optimization of the properties of the barrier, for instance, by introducing spin-selective insulators. Moreover, Hanle measurements in a nonlocal three-terminal configuration serve to confirm the injection of spin-polarized electrons into a CNT. In particular, spin accumulation and spin lifetime extracted from the Hanle effect turn out to be of similar magnitude as in state-of-the-art spin-valve devices based on Si or graphene. Our study provides a deeper understanding of the barrier dependence of the TMR in CNTs, facilitating the development of applications of this one-dimensional material in spintronic devices.

ACKNOWLEDGMENTS

The authors thank S. Trellenkamp for *e*-beam writing, T. Jansen for metallization, and N. Schnitzler, H. Kertz, and H. Pfeifer for technical assistance with the cryostat setups. We thank D. Bürgler, G. Schmidt, and P. Mavropoulos for fruitful discussions. The *wsxM* software is used to analyze all AFM data [84]. M. M. acknowledges financial support from the Alexander von Humboldt Foundation and the Polish Ministry of Science and Higher Education through a young scientist fellowship (0066/E-336/9/2014). C. Meyer acknowledges financial support from the DFG Research unit FOR912.

-
- [1] I. Žutić, J. Fabian, and S. Das Sarma, Spintronics: Fundamentals and applications, *Rev. Mod. Phys.* **76**, 323 (2004).
 - [2] W. Han, R. K. Kawakami, M. Gmitra, and J. Fabian, Graphene spintronics, *Nat. Nanotechnol.* **9**, 794 (2014).
 - [3] M. Drögel, F. Volmer, M. Wolter, B. Terrés, K. Watanabe, T. Taniguchi, G. Güntherodt, C. Stampfer, and B. Beschoten, Nanosecond spin lifetimes in single- and few-layer graphene-hBN heterostructures at room temperature, *Nano Lett.* **14**, 6050 (2014).
 - [4] B. Dlubak, M.-B. Martin, C. Deranlot, B. Servet, S. Xavier, R. Mattana, M. Sprinkle, C. Berger, W. A. DeHeer, F. Petroff, A. Anane, P. Seneor, and A. Fert, Highly efficient spin transport in epitaxial graphene on SiC, *Nat. Phys.* **8**, 557 (2012).
 - [5] N. Tombros, C. Jozsa, M. Popinciuc, H. T. Jonkman, and B. J. van Wees, Electronic spin transport and spin precession in single graphene layers at room temperature, *Nature (London)* **448**, 571 (2007).
 - [6] W. Han, K. Pi, K. M. McCreary, Y. Li, J. J. I. Wong, A. G. Swartz, and R. K. Kawakami, Tunneling Spin Injection into

- Single Layer Graphene, *Phys. Rev. Lett.* **105**, 167202 (2010).
- [7] M. V. Kamalakar, A. Dankert, J. Bergsten, T. Ive, and S. P. Dash, Enhanced tunnel spin injection into graphene using chemical vapor deposited hexagonal boron nitride, *Sci. Rep.* **4**, 6146 (2014).
- [8] T. Kazuhito, W. A. Bruce, and A. Hiroki, Coherent transport of electron spin in a ferromagnetically contacted carbon nanotube, *Nature (London)* **401**, 572 (1999).
- [9] D. Mann, A. Javey, J. Kong, Q. Wang, and H. Dai, Ballistic transport in metallic nanotubes with reliable Pd ohmic contacts, *Nano Lett.* **3**, 1541 (2003).
- [10] B. Zhao, I. Mönch, H. Vinzelberg, T. Mühl, and C. M. Schneider, Spin-coherent transport in ferromagnetically contacted carbon nanotubes, *Appl. Phys. Lett.* **80**, 3144 (2002).
- [11] A. Jensen, J. R. Hauptmann, J. Nygård, and P. E. Lindelof, Magnetoresistance in ferromagnetically contacted single-wall carbon nanotubes, *Phys. Rev. B* **72**, 035419 (2005).
- [12] J. Sann, J. Gramich, A. Baumgartner, M. Weiss, and C. Schönenberger, Optimized fabrication and characterization of carbon nanotube spin valves, *J. Appl. Phys.* **115**, 174309 (2014).
- [13] A. Dirnau, M. Grifoni, A. Prüfling, D. Steininger, A. K. Hüttel, and C. Strunk, Transport across a carbon nanotube quantum dot contacted with ferromagnetic leads: Experiment and nonperturbative modeling, *Phys. Rev. B* **91**, 195402 (2015).
- [14] S. Sahoo, T. Kontos, J. Furer, C. Hoffmann, M. Gräber, A. Cottet, and C. Schönenberger, Electric field control of spin transport, *Nat. Phys.* **1**, 99 (2005).
- [15] H. T. Man, I. J. W. Wever, and A. F. Morpurgo, Spin-dependent quantum interference in single-wall carbon nanotubes with ferromagnetic contacts, *Phys. Rev. B* **73**, 241401 (2006).
- [16] V. Zólyomi, J. Koltai, Á. Ruzsnyák, J. Kürti, Á. Gali, F. Simon, H. Kuzmany, Á. Szabados, and P. R. Surján, Inter-shell interaction in double walled carbon nanotubes: Charge transfer and orbital mixing, *Phys. Rev. B* **77**, 245403 (2008).
- [17] M. Koshino, P. Moon, and Y.-W. Son, Incommensurate double-walled carbon nanotubes as one-dimensional moiré crystals, *Phys. Rev. B* **91**, 035405 (2015).
- [18] J. Kong, H. T. Soh, A. M. Cassell, C. F. Quate, and H. J. Dai, Synthesis of individual single-walled carbon nanotubes on patterned silicon wafers, *Nature (London)* **395**, 878 (1998).
- [19] C. Feuillet-Palma, T. Delattre, P. Morfin, J.-M. Berroir, G. Fève, D. C. Glatli, B. Plaçais, A. Cottet, and T. Kontos, Conserved spin and orbital phase along carbon nanotubes connected with multiple ferromagnetic contacts, *Phys. Rev. B* **81**, 115414 (2010).
- [20] D. Preusche, S. Schmidmeier, E. Pallecchi, C. Dietrich, A. K. Hüttel, J. Zweck, and C. Strunk, Characterization of ferromagnetic contacts to carbon nanotubes, *J. Appl. Phys.* **106**, 084314 (2009).
- [21] C. Morgan, C. M. Schneider, and C. Meyer, Permalloy and Co₅₀Pd₅₀ as ferromagnetic contacts for magnetoresistance measurements in carbon nanotube-based transport structures, *J. Appl. Phys.* **111**, 07B309 (2012).
- [22] C. Morgan, K. Schmalbuch, F. García-Sánchez, C. M. Schneider, and C. Meyer, Structure and magnetization in CoPd thin films and nanocontacts, *J. Magn. Magn. Mater.* **325**, 112 (2013).
- [23] C. Spudat, C. Meyer, K. Goss, and C. M. Schneider, Peapod synthesis depending on the number of nanotube sidewalls, *Phys. Status Solidi B* **246**, 2498 (2009).
- [24] E. I. Rashba, Theory of electrical spin injection: Tunnel contacts as a solution of the conductivity mismatch problem, *Phys. Rev. B* **62**, R16267 (2000).
- [25] K. Tsukagoshi, N. Yoneya, S. Uryu, Y. Aoyagi, A. Kanda, Y. Ootuka, and B. W. Alphenaar, Carbon nanotube devices for nanoelectronics, *Physica (Amsterdam)* **323B**, 107 (2002).
- [26] C. Morgan, D. Metten, C. M. Schneider, and C. Meyer, Effect of contact geometry on magnetoresistance in CoPd-contacted carbon nanotubes, *Phys. Status Solidi B* **250**, 2622 (2013).
- [27] K. K. Likharev, Single-electron devices and their applications, *Proc. IEEE* **87**, 606 (1999).
- [28] The “jump” that appears between the transition regime and the FN regime is caused by an internal switching of measurement ranges in the Keithley 2636B at a current of $I = 1$ nA. It does not affect the qualitative shape of the curve (cf. Fig. 1).
- [29] M. Lenzlinger and E. H. Snow, Fowler-Nordheim tunneling into thermally grown SiO₂, *J. Appl. Phys.* **40**, 278 (1969).
- [30] M. Müller, G.-X. Miao, and J. S. Moodera, Exchange splitting and bias-dependent transport in EuO spin filter tunnel barriers, *Europhys. Lett.* **88**, 47006 (2009).
- [31] J. S. Moodera, L. R. Kinder, T. M. Wong, and R. Meservey, Large Magnetoresistance at Room Temperature in Ferromagnetic Thin Film Tunnel Junctions, *Phys. Rev. Lett.* **74**, 3273 (1995).
- [32] T. Miyazaki and N. Tezuka, Giant magnetic tunneling effect in Fe/Al₂O₃/Fe junction, *J. Magn. Magn. Mater.* **139**, L231 (1995).
- [33] D. D. Jayaprawira, K. Tsunekawa, M. Nagai, H. Maehara, S. Yamagata, N. Watanabe, S. Yuasa, Y. Suzuki, and K. Ando, 230% room-temperature magnetoresistance in CoFeB/MgO/CoFeB magnetic tunnel junctions, *Appl. Phys. Lett.* **86**, 092502 (2005).
- [34] X. Zhang, S. Mizukami, T. Kubota, Q. Ma, M. Oogane, H. Naganuma, Y. Ando, and T. Miyazaki, Observation of a large spin-dependent transport length in organic spin valves at room temperature, *Nat. Commun.* **4**, 1392 (2013).
- [35] M. Bowen, A. Barthélémy, V. Bellini, M. Bibes, P. Seneor, E. Jacquet, J.-P. Contour, and P. H. Dederichs, Observation of Fowler-Nordheim hole tunneling across an electron tunnel junction due to total symmetry filtering, *Phys. Rev. B* **73**, 140408 (2006).
- [36] T. L. Qu, J. Li, Y. G. Zhao, J. W. Mei, X. Liu, H. F. Tian, J. P. Shi, S. M. Guo, J. Li, D. N. Zheng, and J. Q. Li, Nonlinear current-voltage behavior and giant positive magnetoresistance in nonmagnetic Au/Yttria-stabilized zirconia/Si heterostructures, *Appl. Phys. Lett.* **95**, 242113 (2009).
- [37] T. Nagahama, T. S. Santos, and J. S. Moodera, Enhanced Magnetotransport at High Bias in Quasimagnetic Tunnel Junctions with EuS Spin-Filter Barriers, *Phys. Rev. Lett.* **99**, 016602 (2007).
- [38] W. Kim, A. Javey, R. Tu, J. Cao, Q. Wang, and H. J. Dai, Electrical contacts to carbon nanotubes down to 1 nm in diameter, *Appl. Phys. Lett.* **87**, 173101 (2005).

- [39] M. Bockrath, D. H. Cobden, P. L. McEuen, N. G. Chopra, A. Zettl, A. Thess, and R. E. Smalley, Single-electron transport in ropes of carbon nanotubes, *Science* **275**, 1922 (1997).
- [40] N. Nemec, D. Tománek, and G. Cuniberti, Contact Dependence of Carrier Injection in Carbon Nanotubes: An *Ab Initio* Study, *Phys. Rev. Lett.* **96**, 076802 (2006).
- [41] B. Shan and K. Cho, *Ab initio* study of Schottky barriers at metal-nanotube contacts, *Phys. Rev. B* **70**, 233405 (2004).
- [42] J. C. Slonczewski, Conductance and exchange coupling of two ferromagnets separated by a tunneling barrier, *Phys. Rev. B* **39**, 6995 (1989).
- [43] C. L. Kane and E. J. Mele, Size, Shape, and Low Energy Electronic Structure of Carbon Nanotubes, *Phys. Rev. Lett.* **78**, 1932 (1997).
- [44] R. Egger and A. O. Gogolin, Correlated transport and non-fermi-liquid behavior in single-wall carbon nanotubes, *Eur. Phys. J. B* **3**, 281 (1998).
- [45] A. Cottet, T. Kontos, S. Sahoo, H. T. Man, M.-S. Choi, W. Belzig, C. Bruder, A. F. Morpurgo, and C. Schönenberger, Nanospintronics with carbon nanotubes, *Semicond. Sci. Technol.* **21**, S78 (2006).
- [46] W. Liang, M. Bockrath, D. Bozovic, J. H. Hafner, M. Tinkham, and H. Park, Fabry-Pérot interference in a nanotube electron waveguide, *Nature (London)* **411**, 665 (2001).
- [47] H. T. Man and A. F. Morpurgo, Sample-Specific and Ensemble-Averaged Magnetoconductance of Individual Single-Wall Carbon Nanotubes, *Phys. Rev. Lett.* **95**, 026801 (2005).
- [48] A. Cottet, T. Kontos, W. Belzig, C. Schönenberger, and C. Bruder, Controlling spin in an electronic interferometer with spin-active interfaces, *Europhys. Lett.* **74**, 320 (2006).
- [49] G. E. Blonder, M. Tinkham, and T. M. Klapwijk, Transition from metallic to tunneling regimes in superconducting microconstrictions: Excess current, charge imbalance, and supercurrent conversion, *Phys. Rev. B* **25**, 4515 (1982).
- [50] Y. Qi, D. Y. Xing, and J. Dong, Relation between Julliere and Slonczewski models of tunneling magnetoresistance, *Phys. Rev. B* **58**, 2783 (1998).
- [51] D. Grundler, Oscillatory Spin-Filtering due to Gate Control of Spin-Dependent Interface Conductance, *Phys. Rev. Lett.* **86**, 1058 (2001).
- [52] C.-M. Hu and T. Matsuyama, Spin Injection across a Heterojunction: A Ballistic Picture, *Phys. Rev. Lett.* **87**, 066803 (2001).
- [53] S. Datta, *Electronic Transport in Mesoscopic Systems* (Cambridge University Press, Cambridge, England, 1997).
- [54] Ya. M. Blanter and M. Büttiker, Shot noise in mesoscopic conductors, *Phys. Rep.* **336**, 1 (2000).
- [55] P. L. McEuen, M. Bockrath, D. H. Cobden, Y.-G. Yoon, and S. G. Louie, Disorder, Pseudospins, and Backscattering in Carbon Nanotubes, *Phys. Rev. Lett.* **83**, 5098 (1999).
- [56] M. Y. Veillette, C. Bena, and L. Balents, Spin precession and oscillations in mesoscopic systems, *Phys. Rev. B* **69**, 075319 (2004).
- [57] M. Misiorny and C. Meyer (to be published).
- [58] P. M. Tedrow and R. Meservey, Spin polarization of electrons tunneling from films of Fe, Co, Ni, and Gd, *Phys. Rev. B* **7**, 318 (1973).
- [59] Z.-S. Shan, Ping He, C. Moore, John Woollam, and D. J. Sellmyer, Behavior of disordered Co-Pd, Co-Ag, and Co-Mo alloys in multilayer interfaces, *J. Appl. Phys.* **73**, 6057 (1993).
- [60] N. Ranjan, R. Gutiérrez, S. Krompiewski, and G. Cuniberti, Electron transport in carbon nanotube-metal system: Contact effects, *Mol. Phys. Rep.* **40**, 125 (2004).
- [61] S. Yuasa, T. Nagahama, A. Fukushima, Y. Suzuki, and K. Ando, Giant room-temperature magnetoresistance in single-crystal Fe/MgO/Fe magnetic tunnel junctions, *Nat. Mater.* **3**, 868 (2004).
- [62] T. Marukame, T. Ishikawa, T. Taira, K. I. Matsuda, T. Uemura, and M. Yamamoto, Giant oscillations in spin-dependent tunneling resistances as a function of barrier thickness in fully epitaxial magnetic tunnel junctions with a MgO barrier, *Phys. Rev. B* **81**, 134432 (2010).
- [63] W. Wang, E. Liu, M. Kodzuka, H. Sukegawa, M. Wojcik, E. Jedryka, G. H. Wu, K. Inomata, S. Mitani, and K. Hono, Coherent tunneling and giant tunneling magnetoresistance in $\text{Co}_2\text{FeAl}/\text{MgO}/\text{CoFe}$ magnetic tunneling junctions, *Phys. Rev. B* **81**, 140402 (2010).
- [64] W. H. Butler, X.-G. Zhang, T. C. Schulthess, and J. M. MacLaren, Spin-dependent tunneling conductance of $\text{Fe}|\text{MgO}|\text{Fe}$ sandwiches, *Phys. Rev. B* **63**, 054416 (2001).
- [65] H. Aurich, A. Baumgartner, F. Freitag, A. Eichler, J. Trbovic, and C. Schönenberger, Permalloy-based carbon nanotube spin-valve, *Appl. Phys. Lett.* **97**, 153116 (2010).
- [66] S. Zhang and P. M. Levy, Models for magnetoresistance in tunnel junctions, *Eur. Phys. J. B* **10**, 599 (1999).
- [67] P. Mavropoulos, N. Papanikolaou, and P. H. Dederichs, Korrington-Kohn-Rostoker Green-function formalism for ballistic transport, *Phys. Rev. B* **69**, 125104 (2004).
- [68] N. Tombros, S. J. van der Molen, and B. J. van Wees, Separating spin and charge transport in single-wall carbon nanotubes, *Phys. Rev. B* **73**, 233403 (2006).
- [69] K. Ono, H. Shimada, and Y. Ootuka, Spin polarization and magneto-Coulomb oscillations in ferromagnetic single electron devices, *J. Phys. Soc. Jpn.* **67**, 2852 (1998).
- [70] H. Shimada, K. Ono, and Y. Ootuka, Driving the single-electron device with a magnetic field, *J. Appl. Phys.* **93**, 8259 (2003).
- [71] S. J. van der Molen, N. Tombros, and B. J. van Wees, Magneto-coulomb effect in spin-valve devices, *Phys. Rev. B* **73**, 220406 (2006).
- [72] C. Gould, C. Rüster, T. Jungwirth, E. Girgis, G. M. Schott, R. Giraud, K. Brunner, G. Schmidt, and L. W. Molenkamp, Tunneling Anisotropic Magnetoresistance: A Spin-Valve-Like Tunnel Magnetoresistance Using a Single Magnetic Layer, *Phys. Rev. Lett.* **93**, 117203 (2004).
- [73] M. Grünewald, M. Wahler, F. Schumann, M. Michelfeit, C. Gould, R. Schmidt, F. Würthner, G. Schmidt, and L. W. Molenkamp, Tunneling anisotropic magnetoresistance in organic spin valves, *Phys. Rev. B* **84**, 125208 (2011).
- [74] K. Wang, T. L. A. Tran, P. Brinks, J. G. M. Sanderink, T. Bolhuis, W. G. van der Wiel, and M. P. de Jong, Tunneling anisotropic magnetoresistance in $\text{Co}/\text{AlO}_x/\text{Al}$ tunnel junctions with fcc Co (111) electrodes, *Phys. Rev. B* **88**, 054407 (2013).
- [75] S. P. Dash, S. Sharma, R. S. Patel, M. P. de Jong, and R. Jansen, Electrical creation of spin polarization in silicon at room temperature, *Nature (London)* **462**, 491 (2009).

- [76] D. H. Cobden, M. Bockrath, P. L. McEuen, A. G. Rinzler, and R. E. Smalley, Spin Splitting and Even-Odd Effects in Carbon Nanotubes, *Phys. Rev. Lett.* **81**, 681 (1998).
- [77] W. Han and R. K. Kawakami, Spin Relaxation in Single-Layer and Bilayer Graphene, *Phys. Rev. Lett.* **107**, 047207 (2011).
- [78] A. Avsar, T.-Y. Yang, S. Bae, J. Balakrishnan, F. Volmer, M. Jaiswal, Z. Yi, S. R. Ali, G. Güntherodt, B. H. Hong, B. Beschoten, and B. Özyilmaz, Toward wafer scale fabrication of graphene based spin valve devices, *Nano Lett.* **11**, 2363 (2011).
- [79] M. H. D. Guimarães, P. J. Zomer, J. Ingla-Aynés, J. C. Brant, N. Tombros, and B. J. van Wees, Controlling Spin Relaxation in Hexagonal BN-Encapsulated Graphene with a Transverse Electric Field, *Phys. Rev. Lett.* **113**, 086602 (2014).
- [80] V. A. Sydoruk, K. Goß, C. Meyer, M. V. Petrychuk, B. A. Danilchenko, P. Weber, C. Stampfer, J. Li, and S. A. Vitusevich, Low-frequency noise in individual carbon nanotube field-effect transistors with top, side and back gate configurations: Effect of gamma irradiation, *Nanotechnology* **25**, 035703 (2014).
- [81] F. Kuemmeth, S. Ilani, D. C. Ralph, and P. L. McEuen, Coupling of spin and orbital motion of electrons in carbon nanotubes, *Nature (London)* **452**, 448 (2008).
- [82] J. J. Viennot, M. C. Dartiailh, A. Cottet, and T. Kontos, Coherent coupling of a single spin to microwave cavity photons, *Science* **349**, 408 (2015).
- [83] B. Dóra, M. Gulácsi, J. Koltai, V. Zólyomi, J. Kürti, and F. Simon, Electron Spin Resonance Signal of Luttinger Liquids and Single-Wall Carbon Nanotubes, *Phys. Rev. Lett.* **101**, 106408 (2008).
- [84] I. Horcas, R. Fernández, J. M. Gómez-Rodríguez, J. Colchero, J. Gómez-Herrero, and A. M. Baro, WSXM: A software for scanning probe microscopy and a tool for nanotechnology, *Rev. Sci. Instrum.* **78**, 013705 (2007).